

# Microstructural Characterization of Thermal Barrier Coatings Glazed by a High Power Laser

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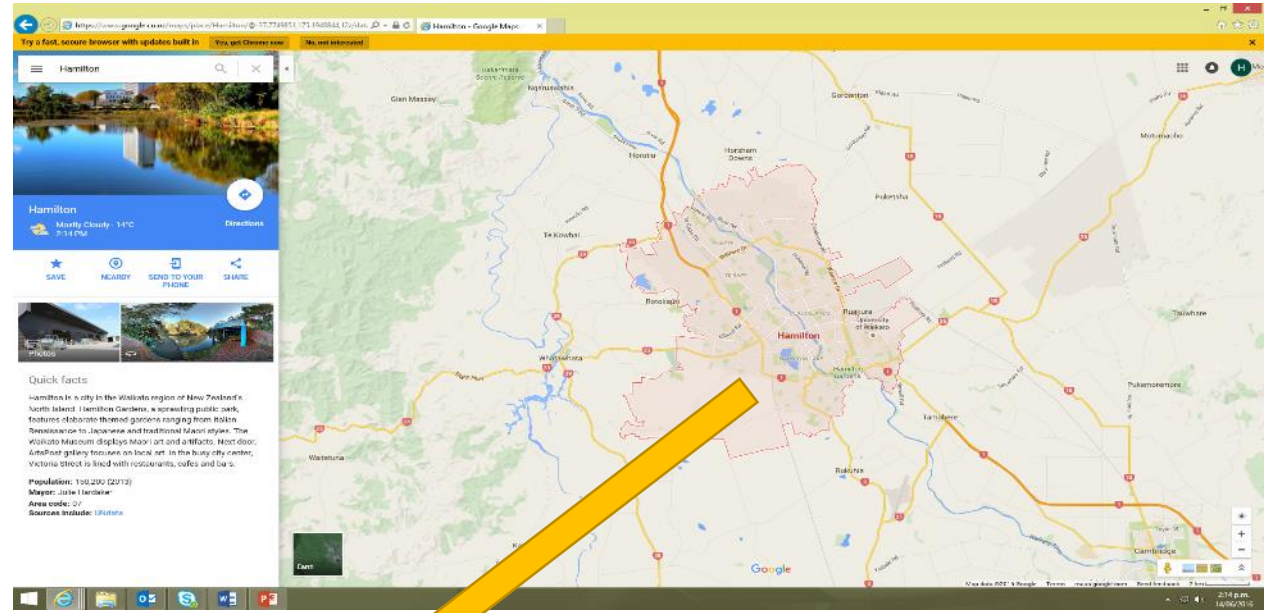
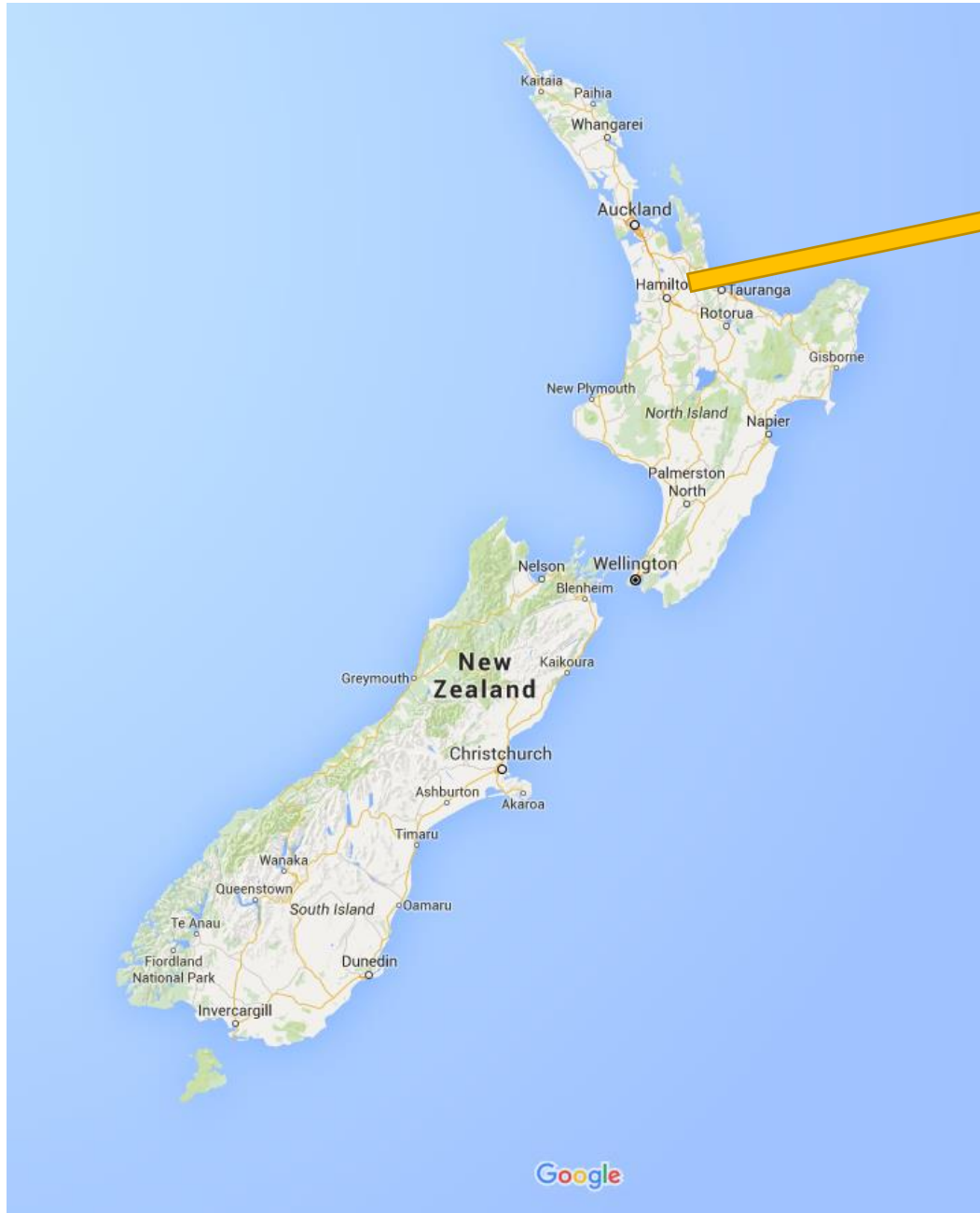
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Plasma-sprayed duplex  $\text{ZrO}_2\text{-8wt.\%Y}_2\text{O}_3$  / MCrAlY (M=Ni and/or Co) TBCs have been widely served in industrial gas turbine engines for metallic components at high temperatures, to

- provide thermal insulation
- improve the efficiency and performance



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Plasma-sprayed TBCs have relatively high interconnected porosity and lamina structure with splat boundaries and microcracks parallel to the interface, which are beneficial for low thermal conductivity.

However, those microstructural features bring out a low bond strength of plasma-sprayed TBCs, lead to a short thermal cycling life.

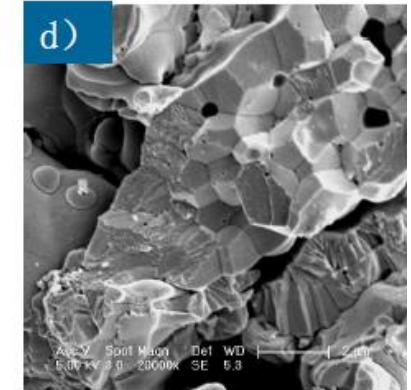
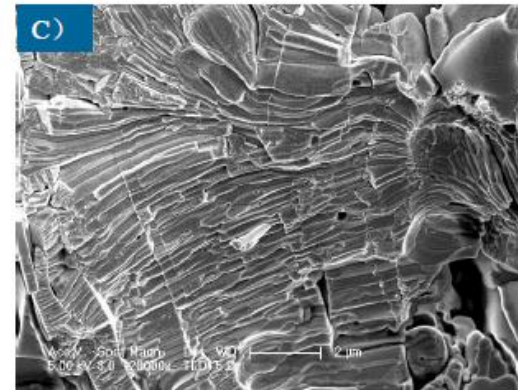
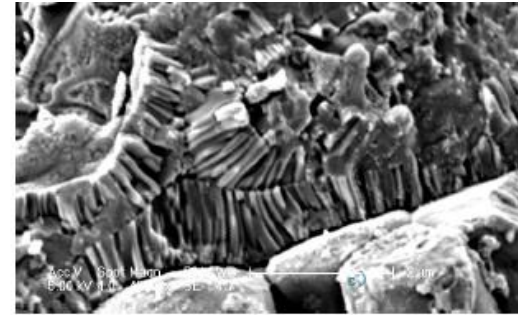


Image @ Hong' s PhD dissertation

Lasers have been used for modification of materials surface since lasers can deliver very high energy density to a localized surface area without significantly heating up the whole body.

Laser-glazing leads to a quick remelting and subsequent rapid solidification of surface, which results in a dense top layer with new microstructure, reduced surface roughness and low porosity

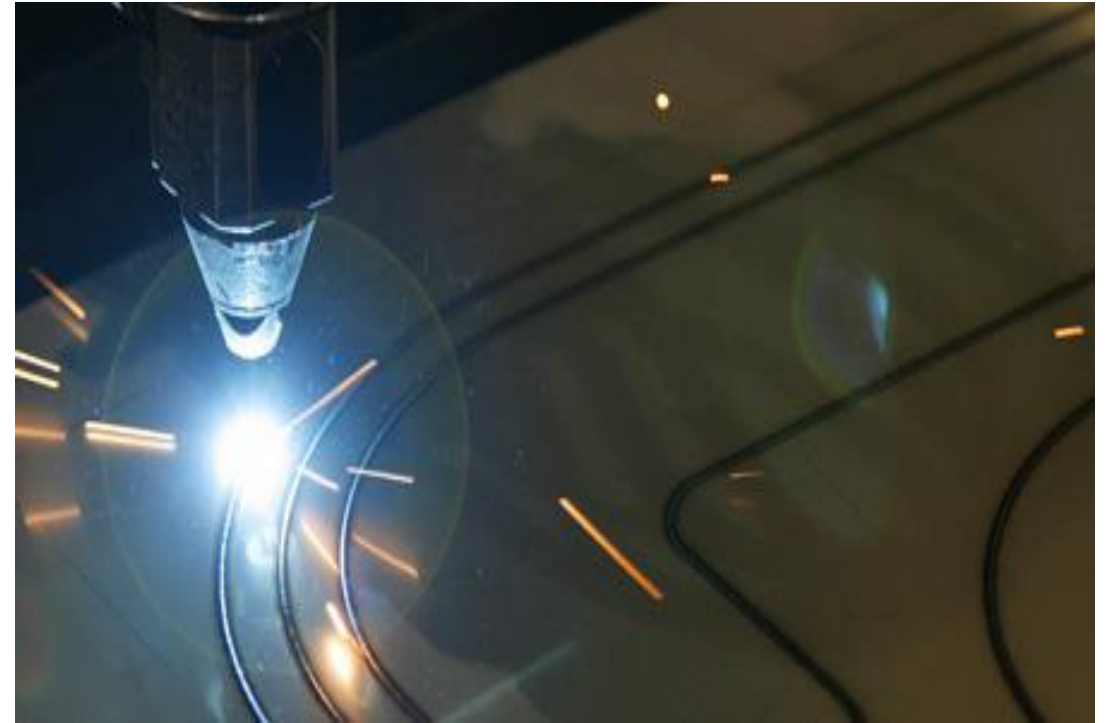


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plasma-sprayed thermal barrier coatings were investigated by glazing with a high power laser in order to modify the structures. A study with focus on the microstructure of the laser-glazed TBCs is reported.



## Experimental Procedure

- **Sample Preparation.**

### **Bond coatings:**

Commercial available NiCrAlY metal powders (HHNiCrAlY-9, CAAMS, Beijing, China) with particle sizes ranging from 10 to 100  $\mu\text{m}$  were air plasma-sprayed to make a coating with thickness of about 0.1 mm

### **Ceramic coatings:**

Commercial 8wt.% Ytria partially stabilized zirconia (YPSZ) powders (HHYPSZ, CAAMS, Beijing, China) with particle sizes from 38.5 to 63  $\mu\text{m}$  were air plasma-sprayed to make a coating with thickness of about 0.3 mm

Air plasma spray system (DH80, CAAMS, Beijing, China). The spraying parameters are listed in Table 1.

Table 1: The spraying parameters for the TBCs

	Current [A]	Voltage [V]	Primary gas Ar [ $l \cdot \text{min}^{-1}$ ]	Primary gas N <sub>2</sub> [ $l \cdot \text{min}^{-1}$ ]	Secondary gas H <sub>2</sub> [ $l \cdot \text{min}^{-1}$ ]	Powder feeding rate [ $g \cdot \text{min}^{-1}$ ]	Stand off distance [mm]
NiCrAlY	480	65	37	-	5	50	100
YPSZ	480	75	-	37	3-5	15	60

- **Laser-glazing.**

The Laser-glazing process was carried out on the plasma-sprayed TBC surface by using an industrial high power (maximum output power of 15 kW) CO<sub>2</sub> continuous wave laser (TLF15000, Triumph Lasercell, Germany) at a power of 3,500 W, as shown in figure 1. The laser system characteristics and processing parameters are listed in table 2.



Fig. 1 The industrial CO<sub>2</sub> continuous wave laser

Table 2 Laser system characteristics and processing parameters

Wave type:	Continuous
Wave length (μm)	10.6
Raw beam size (mm):	25
Focal length (mm):	357
Focal point diameter (mm):	0.7
Defocus distance (mm):	120
Glazing beam diameter (mm):	8.4
Power (W):	3,500
Scanning speed (mm/min):	10000

- **Microstructural Characterization.**

Surface morphologies and coating microstructure were observed by using a field emission scanning electron microscope (FESEM, Model FEI Sirion 200, USA). Fractured and polished cross sections of the laser-glazed ceramic coatings, which were perpendicular to the laser beam travel, were prepared to determine morphological and microstructural modifications along the coating thickness. The arithmetic mean roughnesses ( $R_a$ ) of the ceramic coating surfaces with and without laser-glazing were determined by a mechanical profilometer (TR100, Time Inc., Beijing, China) performed along two orthogonal directions on the coating surfaces.

## Results and Discussion

### Microstructural Analyses

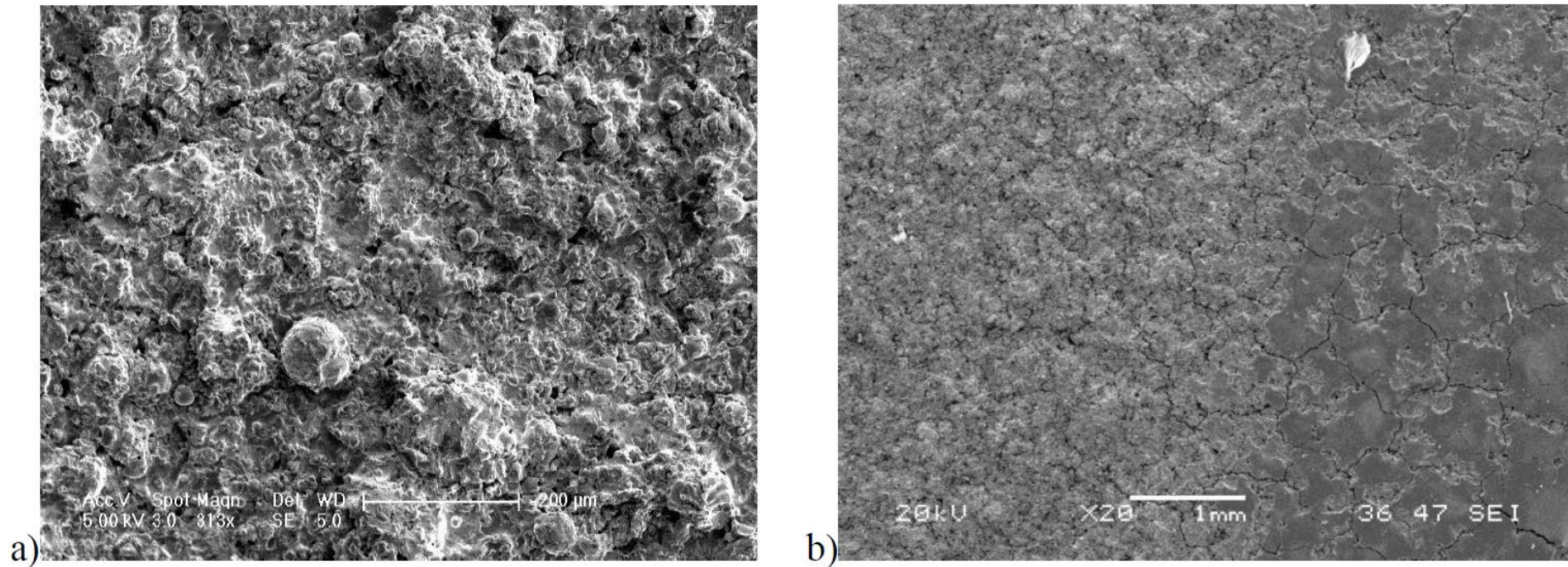


Fig. 2 SEM images of a) the top surface of the as-sprayed ceramic coating and b) the contrast of surface morphology in a region containing both laser treated (right) and untreated (left) areas.

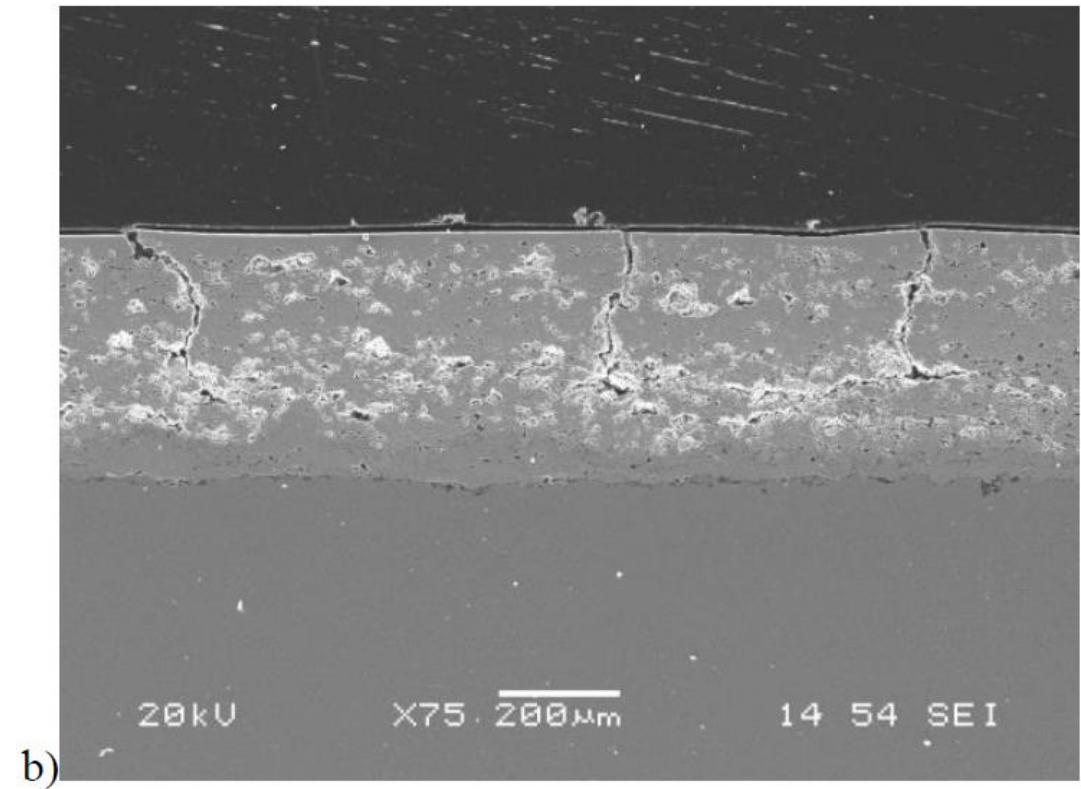
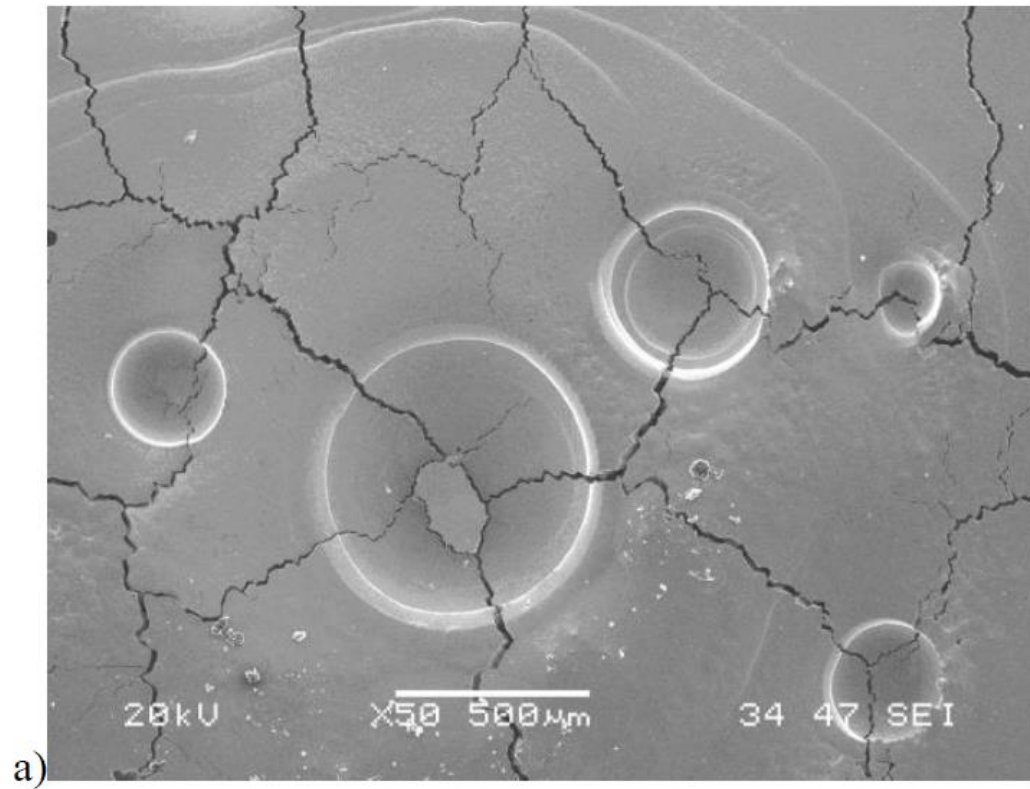
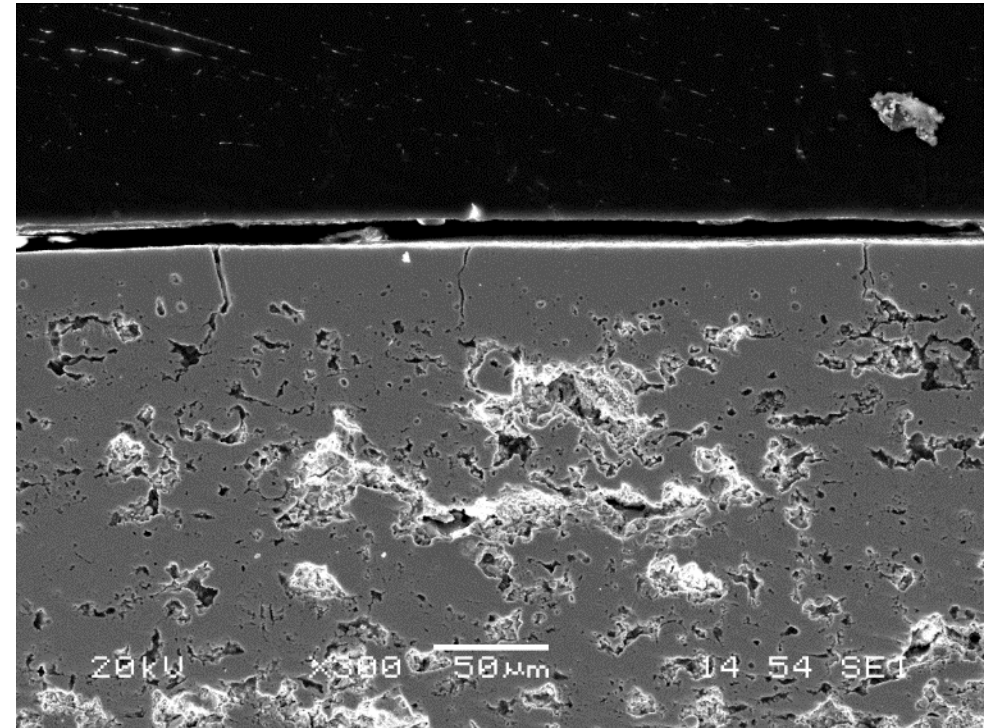
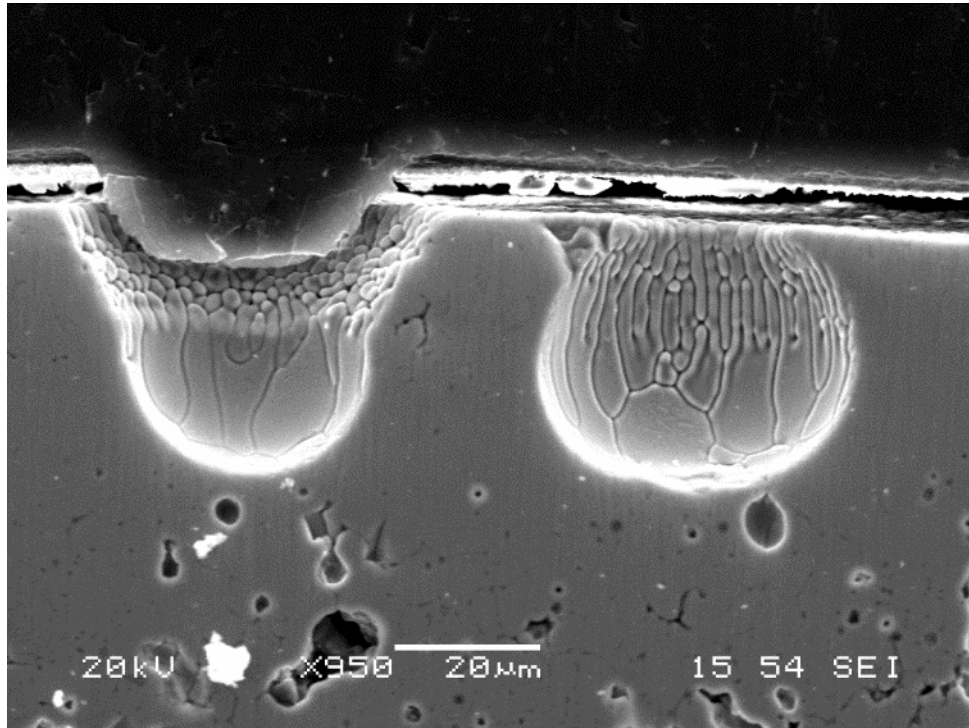


Fig. 3 a) The surface morphology and b) polished cross section of the laser-glazed TBCs



Figures to show the craters and microcracks in the cross sections

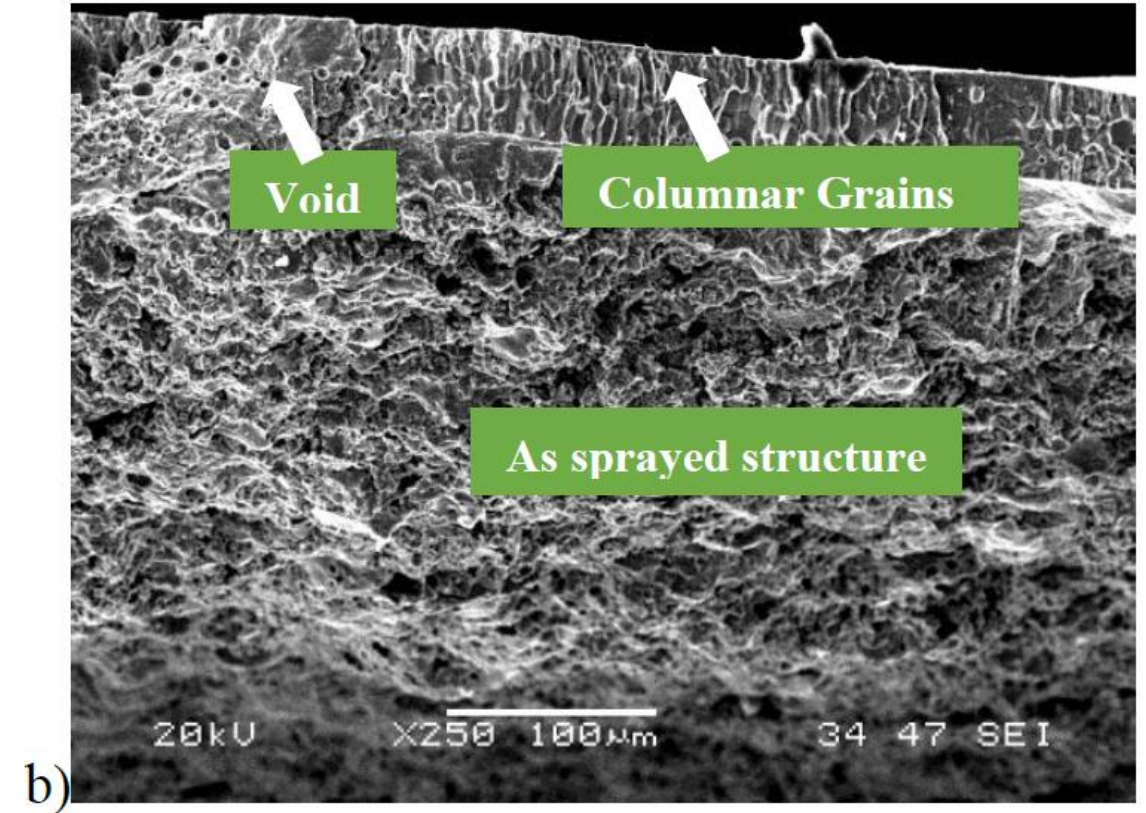
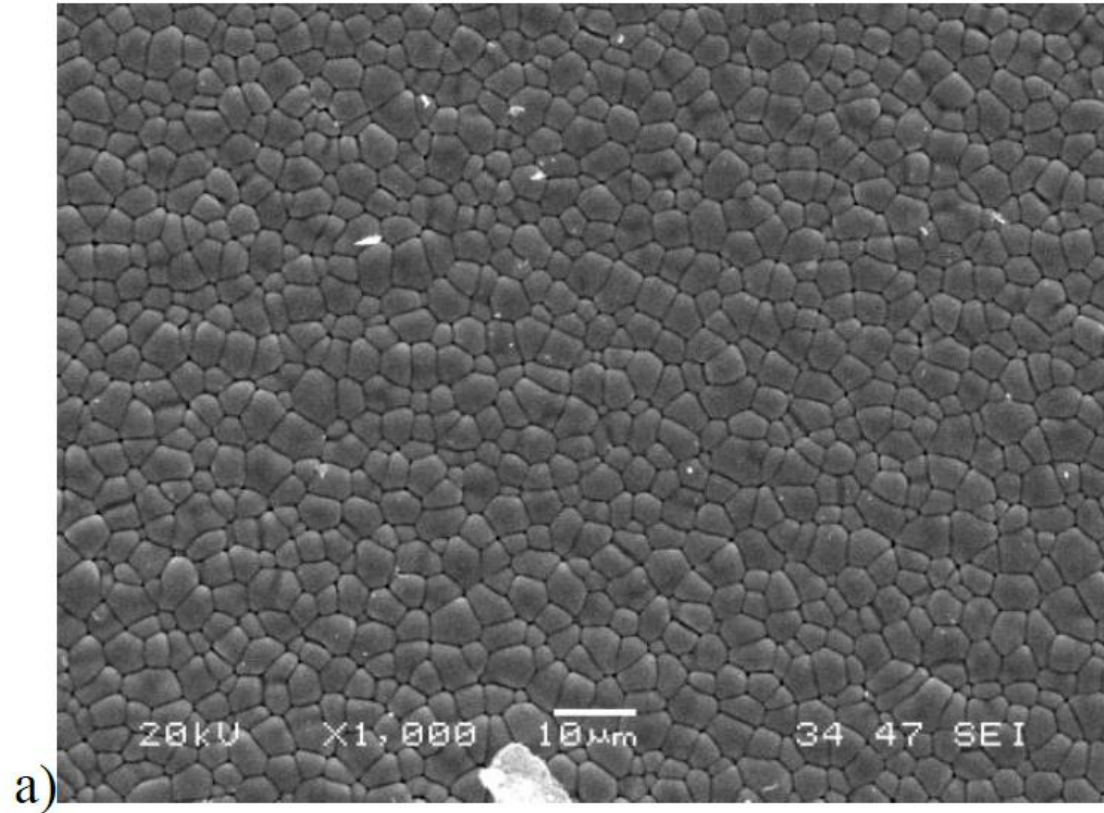
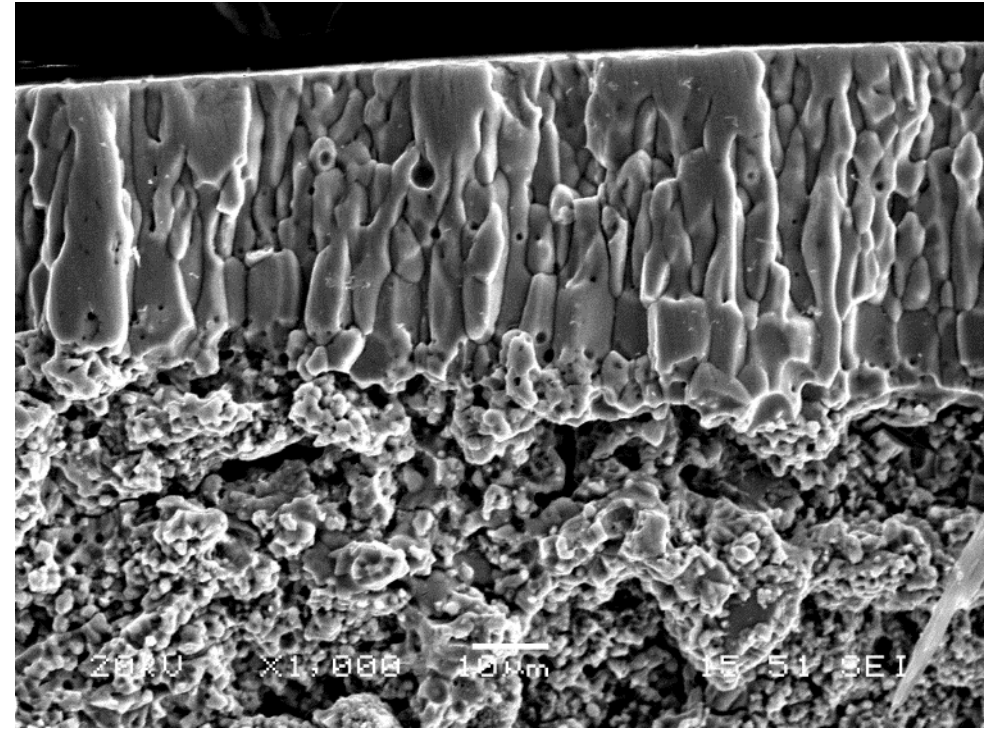
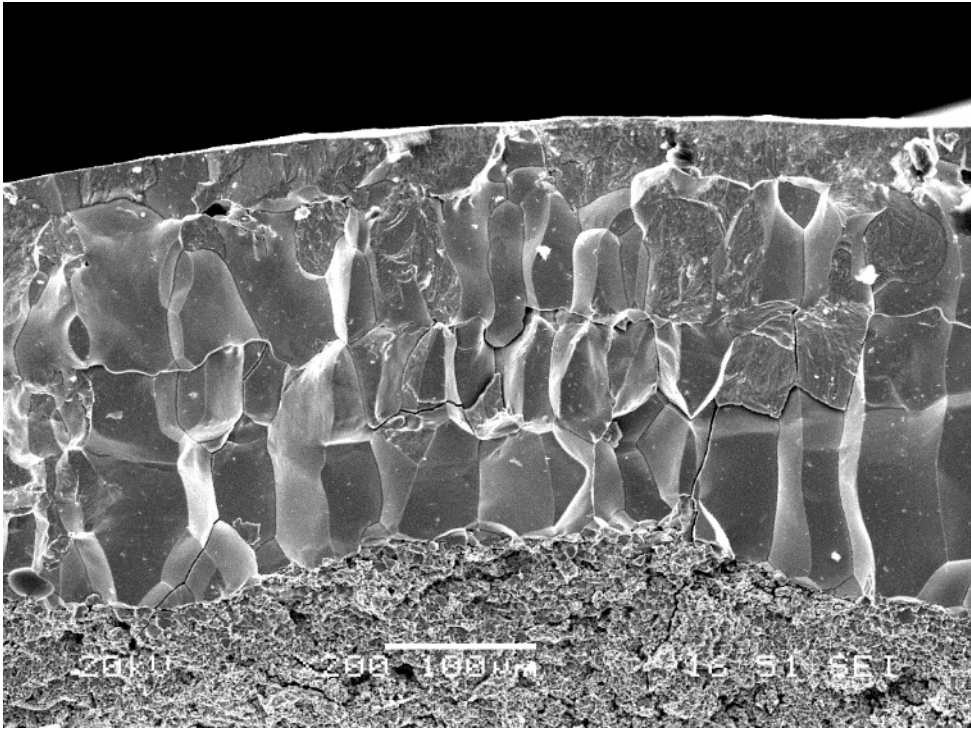


Fig. 4 SEM images of a) the top surface and b) the fractured cross-section of the laser-glazed coatings





Figures to show the fractured cross sections

## Surface Roughness

Surface roughness ( $R_a$ ) has been measured for both specimens. The surface roughness for as-sprayed coating is  $9.3 \pm 0.8 \mu\text{m}$ , while that for the laser-glazed coating is reduced to  $5.1 \pm 0.4 \mu\text{m}$ , as illustrated in Fig. 5. The result indicates an obvious reduction on surface roughness values after laser processing in spite of the forming of craters and microcrack networks on the surface, and thus generates a smoother surface.

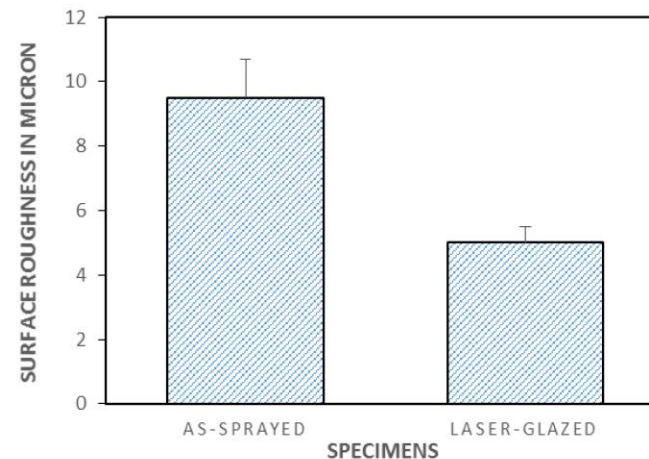


Fig. 5 Comparison of the surface roughness values ( $R_a$ ) between the as-sprayed and the laser-glazed ceramic coatings

# Conclusion

Plasma-sprayed TBCs have been subjected to laser-glazing processes which provide a remelting and subsequent solidification of the surface.

- Laser-glazing carried out on TBCs has resulted in a smooth and dense glazed surface with craters and a network of microcracks.
- The laser-glazed region consists of a columnar microstructure. T
- here are segmentation microcracks in the laser-glazed coatings, which don't run through the coatings along thickness.
- An obvious reduction on surface roughness is achieved after laser processing.

Thanks for your attention